

A microphysical model of rate- and state-friction controlled by dislocation glide and backstress (internal stress) evolution



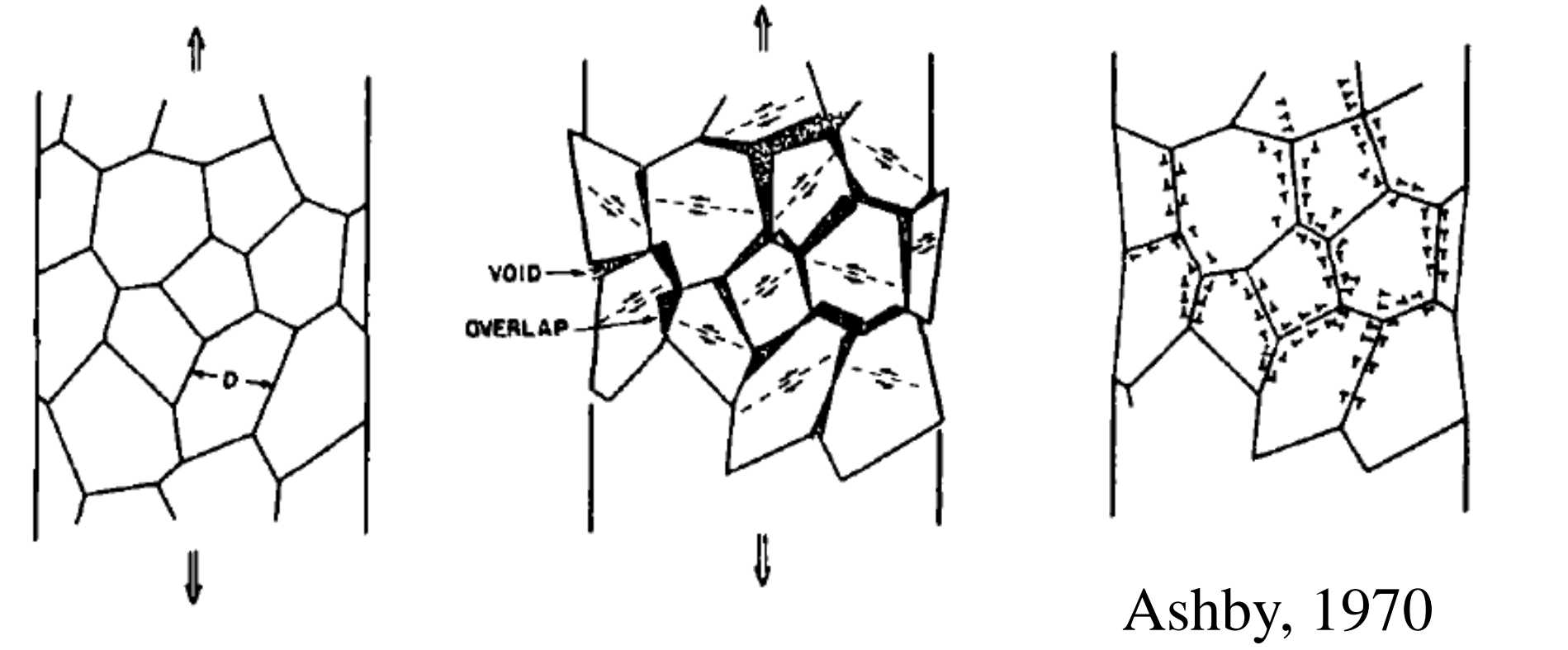
One sentence summary: Internal stress and its temperature dependence controls fault stability and the transition from frictional sliding to bulk deformation.

Christopher A. Thom^{1,2}, Lars N. Hansen³, David L. Goldsby⁴, & Emily E. Brodsky¹

¹University of California, Santa Cruz ²University of Oxford ³University of Minnesota ⁴University of Pennsylvania



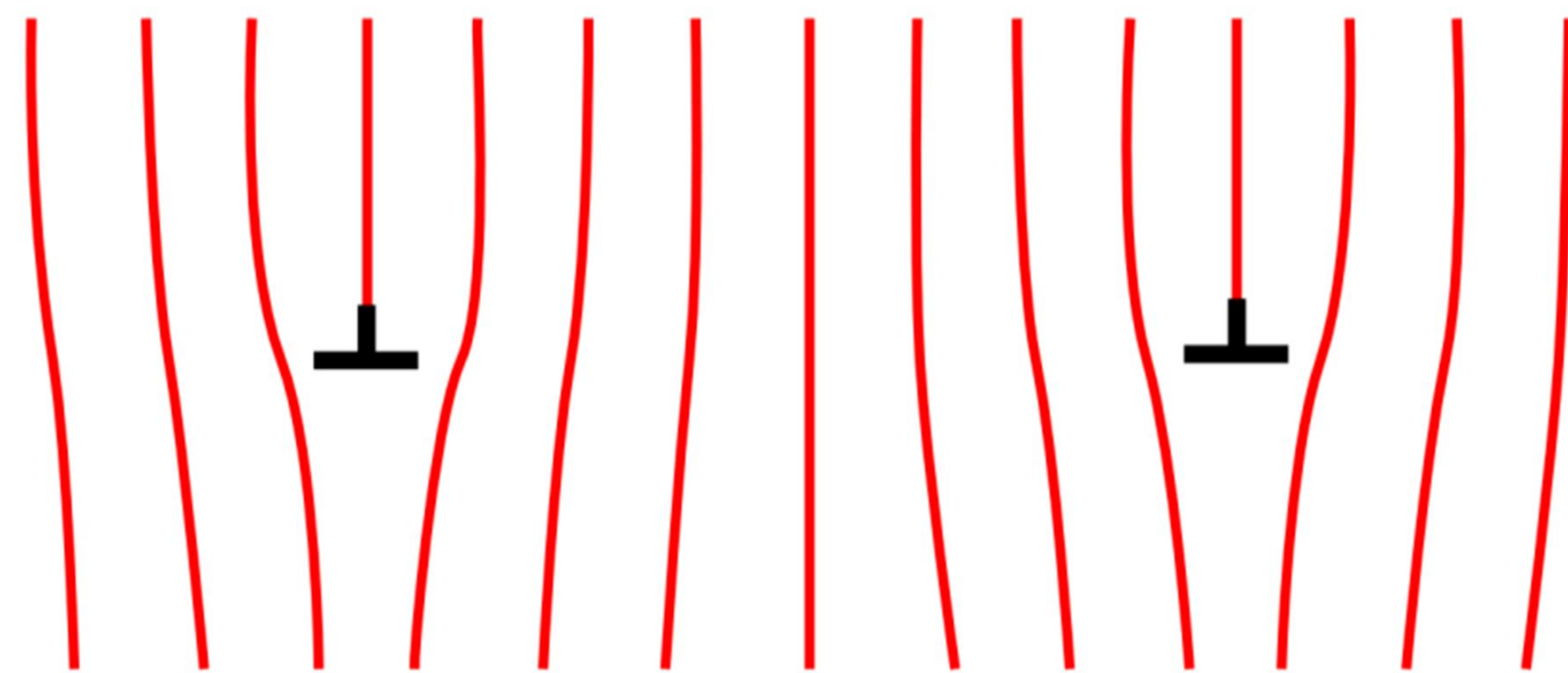
1. What is backstress and what causes it?



Ashby, 1970

When a polycrystalline aggregate or single crystal deforms inhomogeneously, **geometrically necessary dislocations (GNDs)** must be created to 1) maintain compatibility at grain boundaries (above) or 2) **allow for curvature of the crystal lattice** (right).

Because GNDs are dislocations of the same sign, they experience repulsive forces. The **local distortion** of the crystal lattice around GNDs **results in long-range stress fields proportional to GND average spacing** (i.e., dislocation density, see Taylor equation below). In the schematic below, red lines represent lattice planes. If the 2 GNDs were forced closer together, their repulsive stress field would increase.

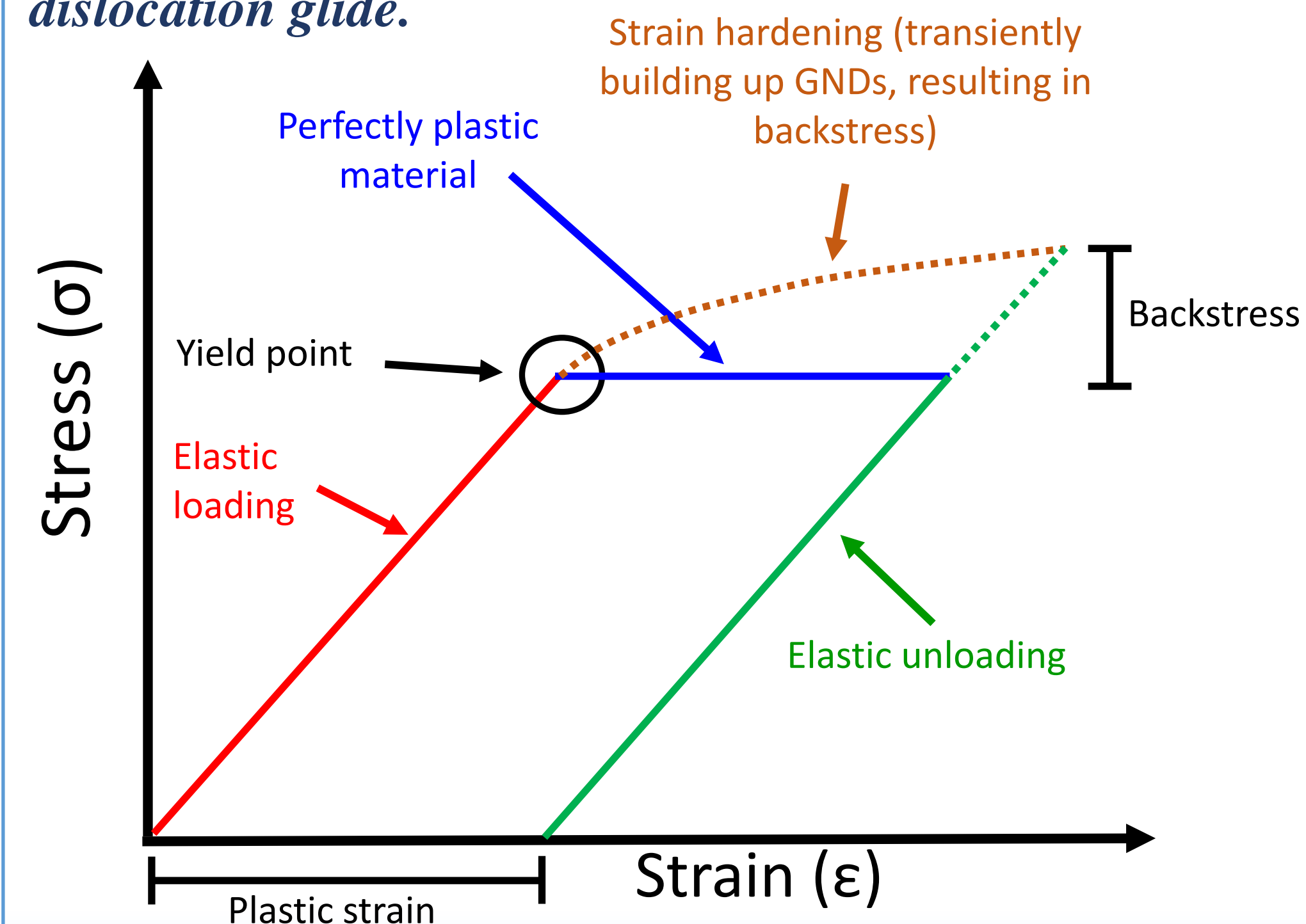


$$\sigma_b = \alpha G b \sqrt{\rho_{GND}}$$

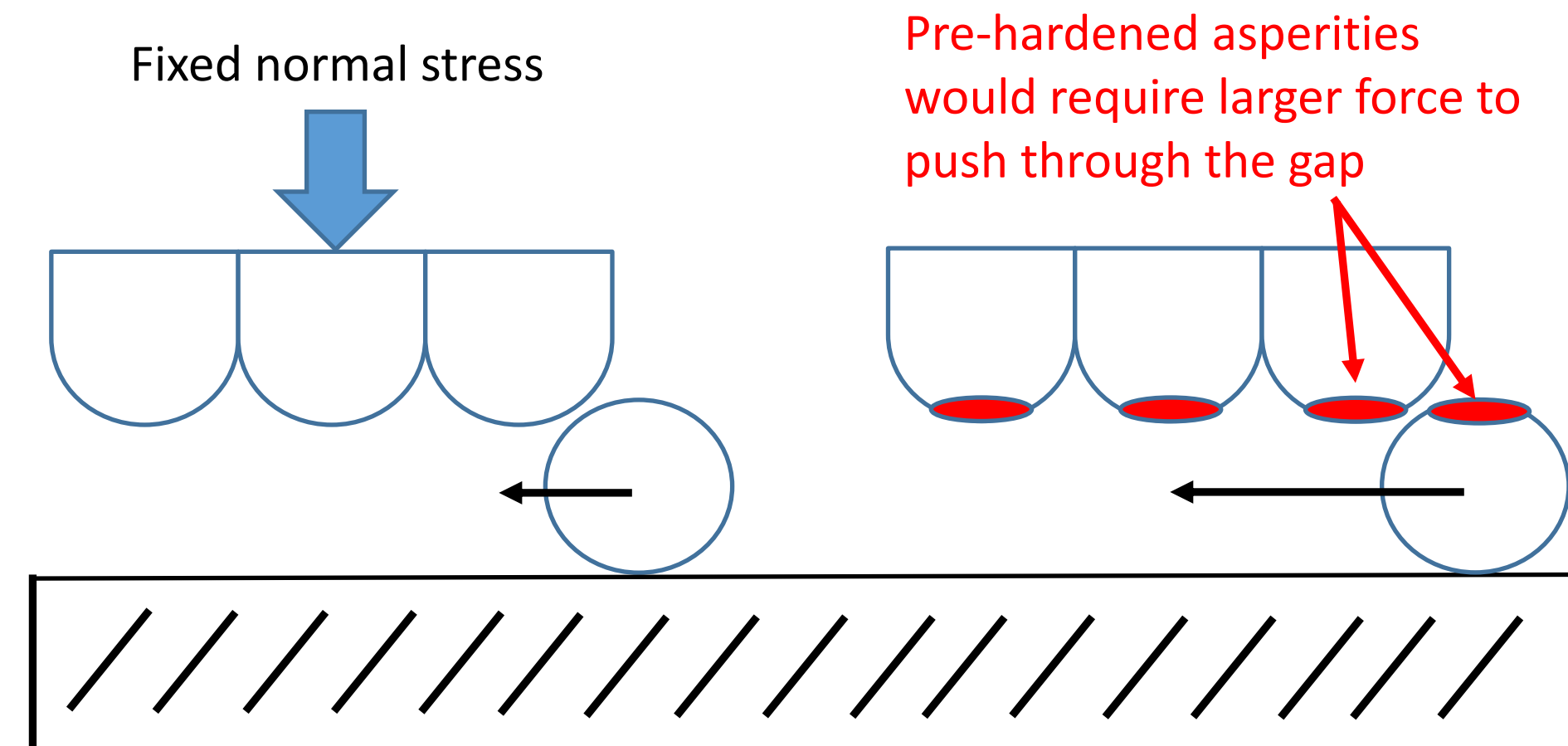
Backstress
Constant
Shear (elastic) modulus

Burgers vector
Dislocation density

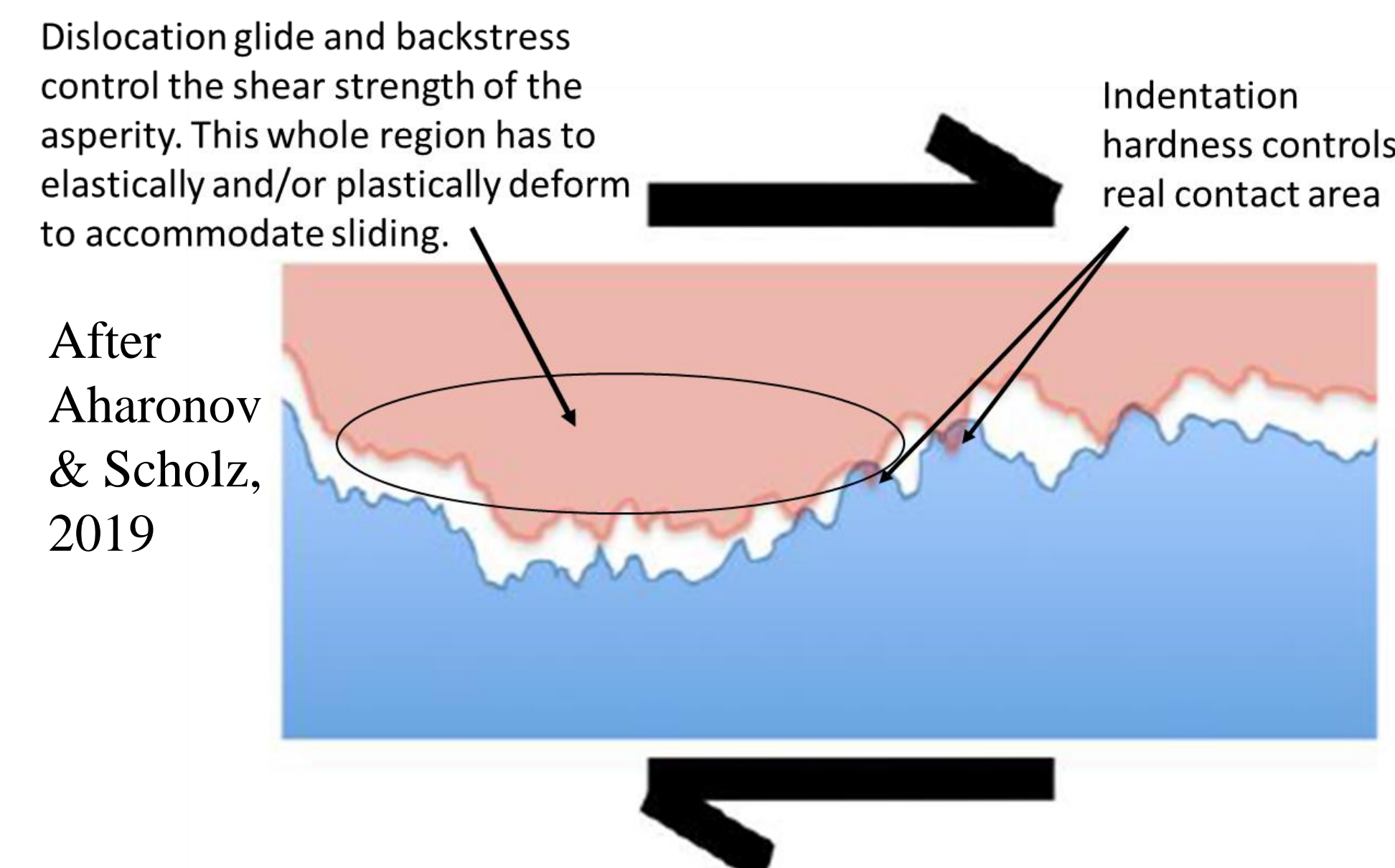
Macroscopically, the accumulation of GNDs during plastic deformation is what causes strain hardening to occur (see schematic below). The total applied stress (at a fixed strain rate) can be written as the sum of two terms (Hansen et al., 2019): 1) the yield stress and 2) the backstress. Thus, **backstress is a stress caused by long-range elastic interactions among lattice dislocations that opposes further deformation by dislocation glide**.



2. How does backstress influence friction?



In the schematic above on the left, a spherical particle is forced through a gap that is bounded by a rigid frictionless plane and a periodic array of 'asperities' with a fixed normal stress. **Some deformation of the particle and the asperity tips must occur** for the particle to slide through the gap. Now consider the same geometry on the right, but with **significant backstresses (internal stresses) locked into the particle and asperities** from previous deformation (shown in red). A **larger force would be required** to move the particle through the gap (i.e., a larger coefficient of friction).



We assume a Bowden & Tabor (1950) description of friction, where the frictional resistance is the average shear stress supported by asperities (from Hansen et al. 2019; 2021 transient plasticity flow law) multiplied by the real contact area of the rough surface (inversely proportional to indentation hardness, H , Thom et al. 2017). Below, we provide rate- and state-friction parameters based on transient creep equations assuming an instantaneous change in the strain rate (a parameter) and the subsequent anelastic response of asperities due to backstress (b parameter). The value of D_c is determined using the Taylor equation (boxed in left panel) and the radius of curvature required to accommodate the density of GNDs (backstress) in the asperities.

$$\mu_0 = \frac{\left(\frac{RT\Sigma}{E} \sinh^{-1} \left[\frac{\dot{\epsilon}_0}{A} \exp \left(\frac{E}{RT} \right) \right] + \sigma_{b,0} \right)}{H} \quad D_c = \frac{2\alpha^2 G^2 b_v}{\sigma_b^2}$$

$$a \approx \frac{\frac{RT\Sigma}{E} \ln \left(\frac{2\dot{\epsilon}_0}{A} \right) + \Sigma + \sigma_{b,0}}{\left(\frac{RT\Sigma}{E} \ln \left(\frac{2\dot{\epsilon}_0}{A} \right) + \Sigma + \sigma_{b,0} \right)} \quad b \approx \frac{(\sigma_{b,max} - \sigma_{b,0})\gamma\beta}{\frac{RT\Sigma}{E} \ln \left(\frac{2\dot{\epsilon}_0}{A} \right) + \Sigma + \sigma_{b,0}}$$

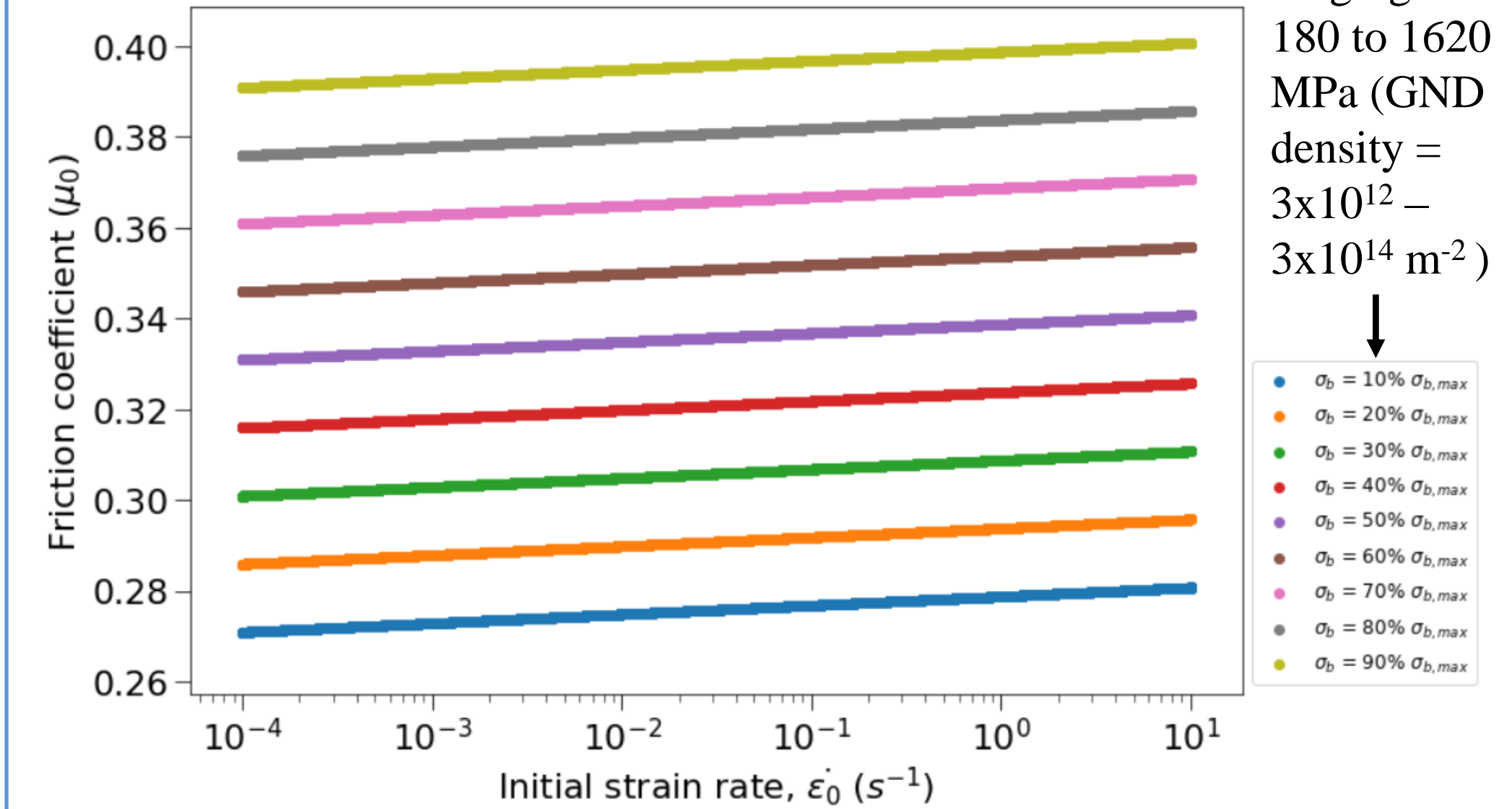
μ_0 friction coefficient
 R gas constant
 T absolute temperature
 Σ Peierls stress
 E activation energy
 $\dot{\epsilon}_0$ strain rate
 σ_b backstress

A material constant
 H indentation hardness
 D_c critical slip distance
 α constant
 G shear modulus
 b_v Burgers vector (lattice spacing)
 a 'rate' parameter

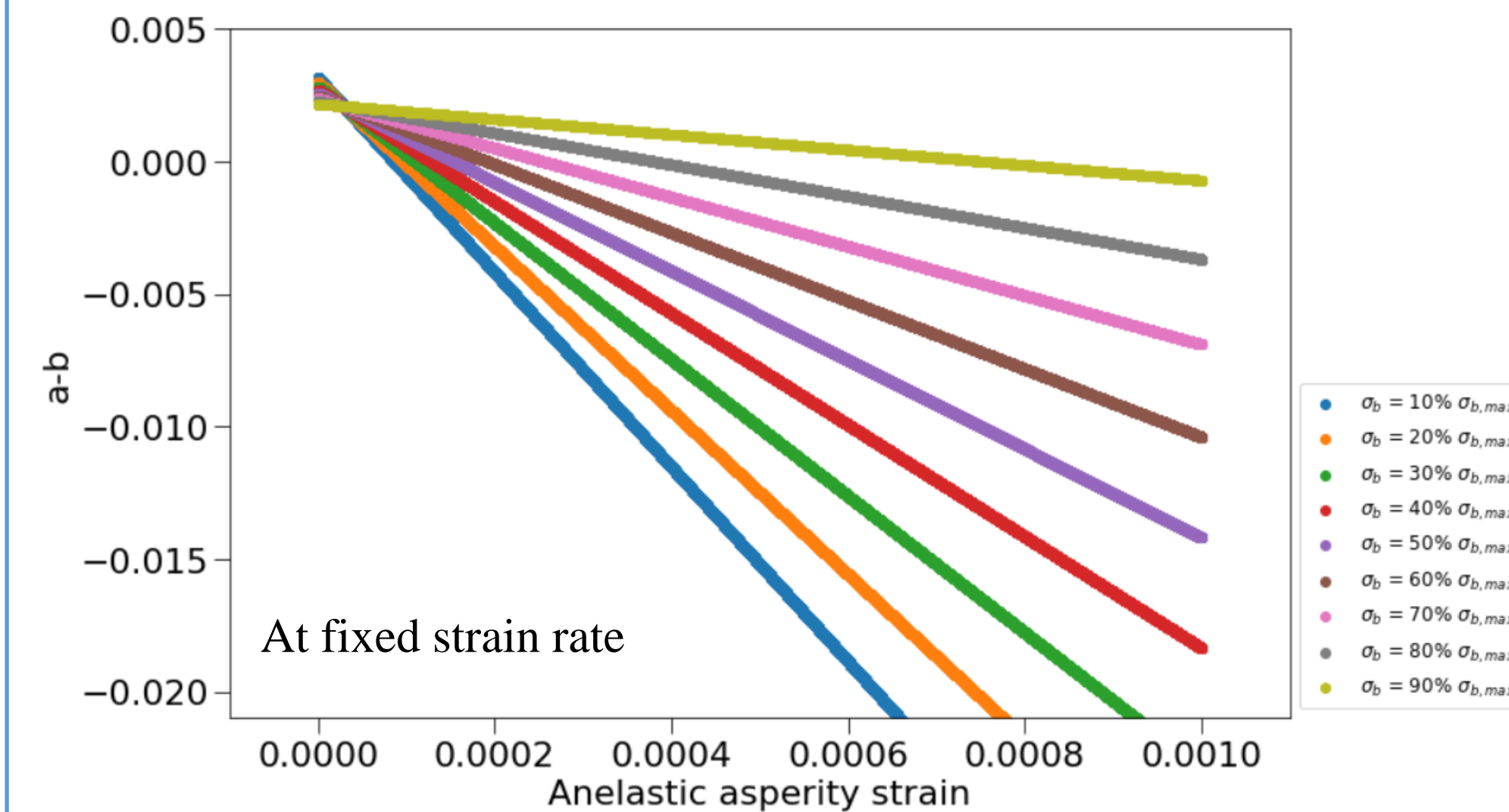
b 'state' parameter
 γ hardening parameter
 β asperity strain

NOTE Backstress evolution equation can be found in Hansen et al. (2021)

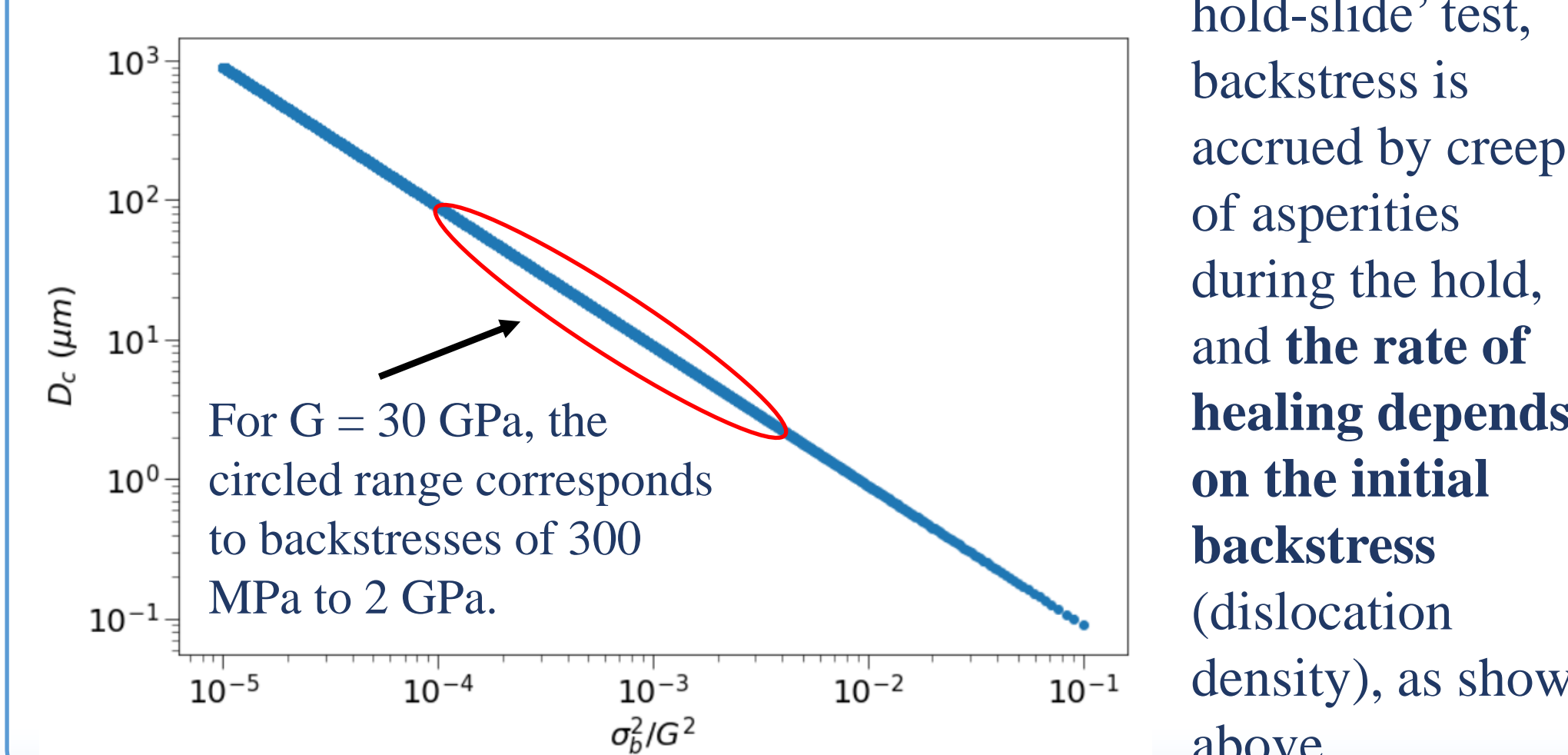
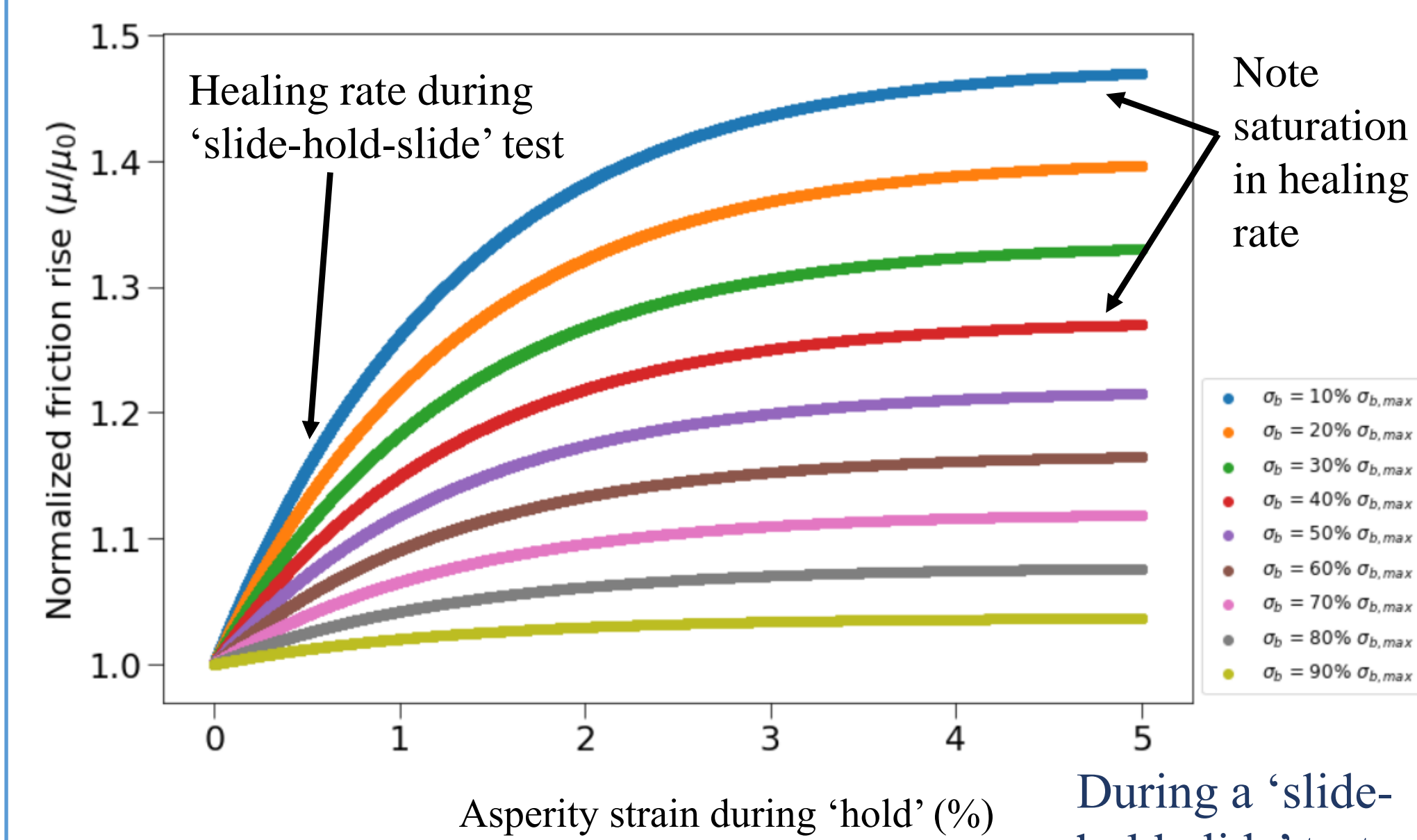
3. Room temperature predictions



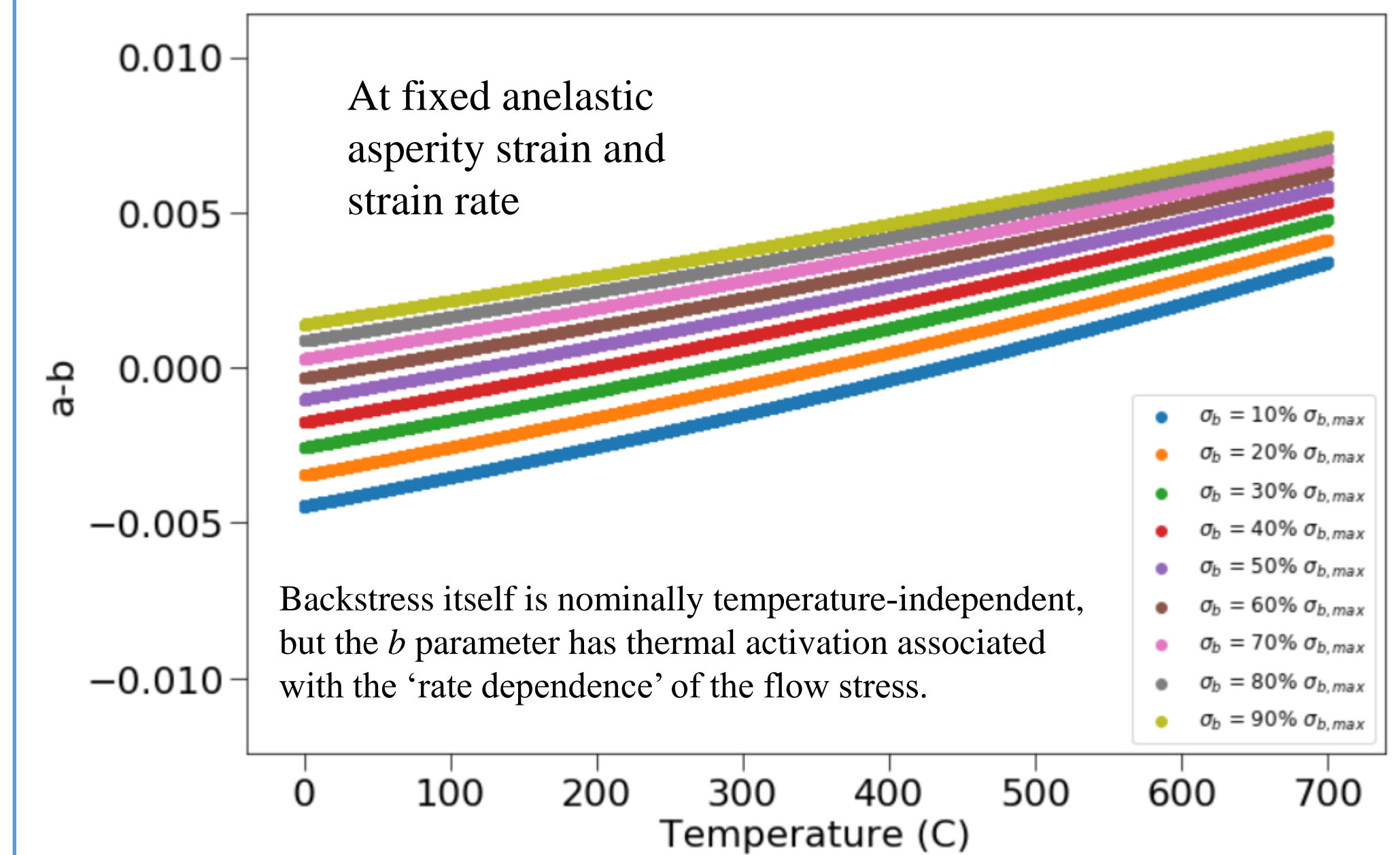
NOTE These predictions at varying levels of backstress (dislocation density) are for olivine (the only mineral with a well-calibrated transient creep flow law), which has a friction coefficient at room temperature of 0.31-0.35 (unpublished experiment, Brown University).



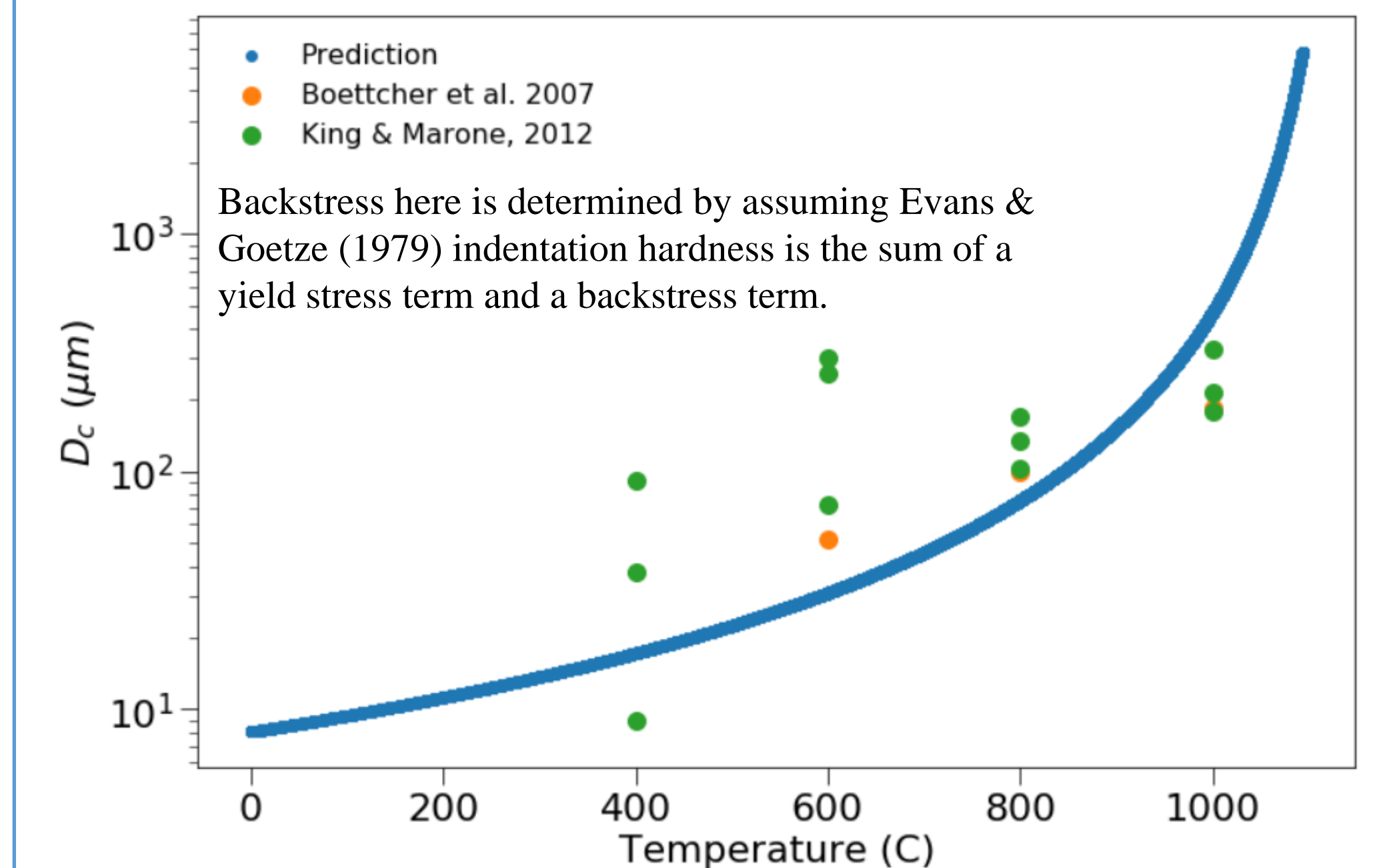
In this framework, the ' a ' value of rate- and state-friction captures the instantaneous 'rate dependence' of friction, as many previous authors have described. However, the ' b ' value is controlled by the evolution of backstress (i.e., dislocation density) within asperities. **$a-b < 0$ can only occur when anelastic relaxation of asperities is more significant than the rate dependence.**



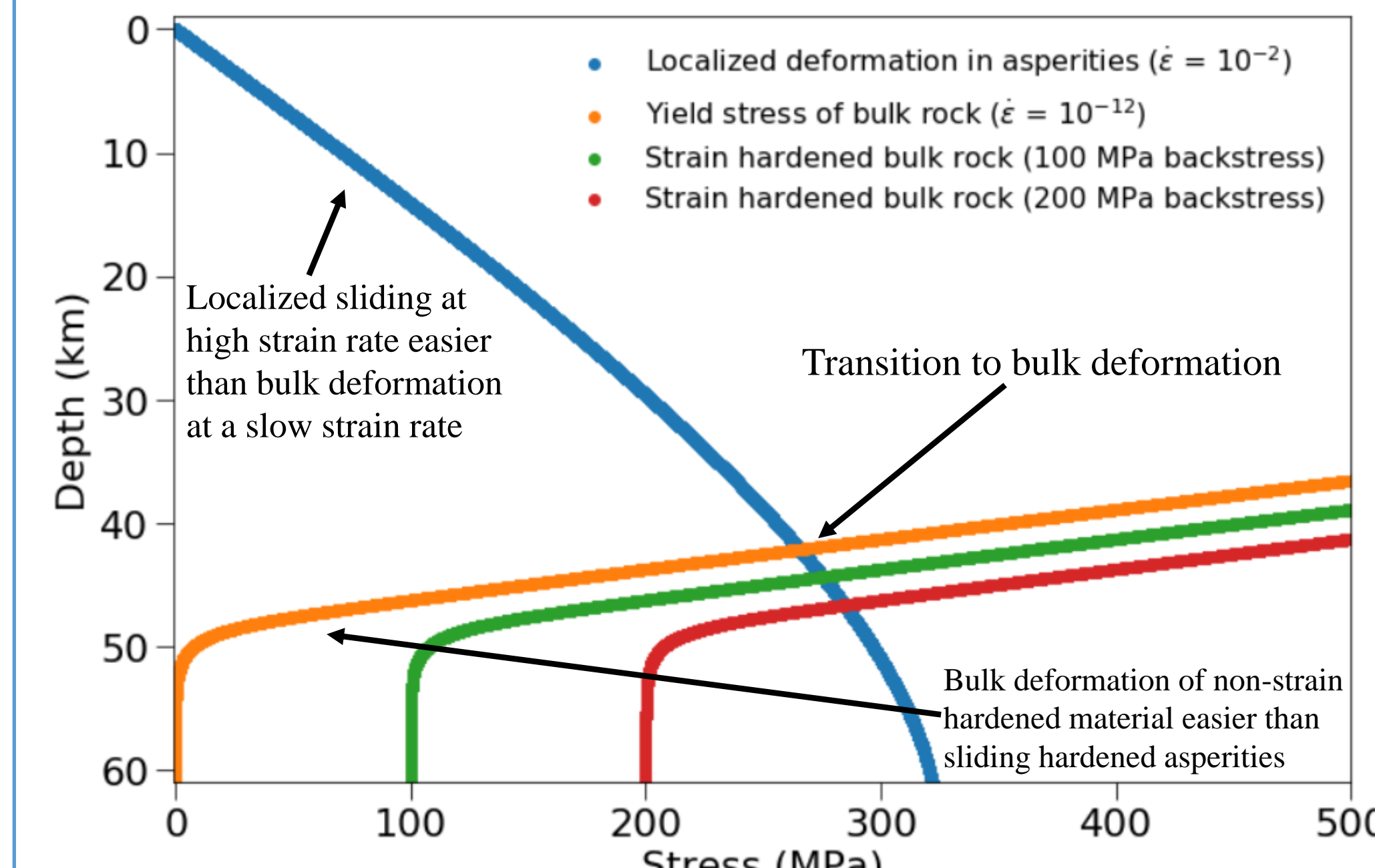
4. Extrapolation to high temperature



Fault stability ($a-b$) is a function of both temperature and backstress (above), and predictions of D_c agree with experimental observations (below).



We predict a major rheological change to occur when highly localized frictional deformation becomes more difficult than deforming bulk rock at geological strain rates:



TAKE HOME MESSAGES

1. We recover rate- and state-frictional behavior using a transient plasticity flow law that accounts for long-range elastic interactions between lattice dislocations.
2. The 'state' dependence is caused by the anelastic response of asperities to large local backstresses (internal stresses).
3. Fault stability ($a-b$ value) exhibits a dependence on both backstress and temperature.